

Monolayer WS₂ Lateral Homosuperlattices with Two-dimensional Periodic Localized Photoluminescence

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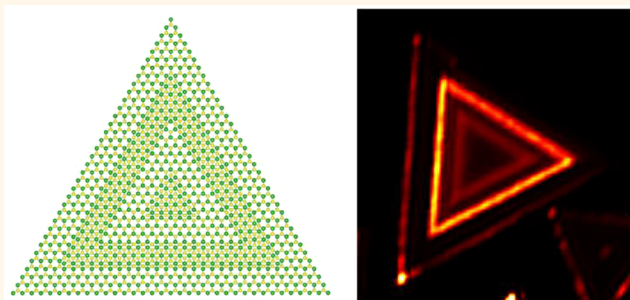
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Supporting Information

ABSTRACT: Homojunctions and homosuperlattices are essential structures and have been widely explored for use in advanced electronic and optoelectronic devices. However, artificially manipulating crystalline phases in two-dimensional (2D) monolayers is still challenging, especially when attempting to engineer lateral homogeneous junctions in a single monolayer of transition metal dichalcogenides (TMDs). Herein, we demonstrate a lateral homosuperlattice (MLHS) with alternating 1T and 2H domains in a 2D WS₂ monolayer plane. In MLHSs, the 2H domains, which are laterally localized and isolated by potential wells, manifest junction interfaces and irradiated photoluminescence (PL) with a lateral periodic distribution in the two-dimensional plane. The studies on MLHSs here can provide further understanding of lateral homojunctions and homosuperlattices in a monolayer plane, providing an alternative route to modulate optical and electronic behaviors in TMD monolayers.

KEYWORDS: WS₂, homosuperlattices, 1T and 2H phases, potential wells, photoluminescence



Homojunctions, heterojunctions, and correlated superlattices are fundamental elements in modern photonics and electronics.^{1–4} Dense integration, obtaining superlattice performance, and downscaling remain pressing challenges for homo/heterojunction devices. Recently, 2D homo/heterostructures (e.g., graphene, TMDs) have shown great potential in the fields of physical research and advanced optoelectronic devices.^{2–10} Specifically, artificial manipulations of the crystalline phases and superlattices of 2D materials have been successfully exploited to produce devices with better integration^{10,11} and performance.^{1–4,7–9,12–16} TMDs stacking structures have been used to realize high carrier mobility and rapid on/off response in functional materials and devices. However, in spite of the progress made, it is still challenging and of the utmost necessity to control the integration precisely, inner interfaces, and band offsets of lateral and vertical homogeneous structures.^{2,7,13} Aiming at these goals, many efforts have been made on thickness modulation, doping, and phase engineering^{13,14,17–21} to realize lateral homojunctions,²² composition-dependent phase variations,²³ and ohmic homomorphic contacts.²⁴

Phase engineering is a well-known and robust way to effectively modulate the lattices and physical properties (e.g., band structure and phonon vibration) of homostructural materials.^{25,26} Nevertheless, it is still urgent in the 2D community to understand and explore 2D lateral homojunctions and superlattices on the horizontal plane. Existing strategies, including ion intercalation,¹³ laser irradiation^{18,24} and electron beam induction,¹⁹ have been used to engineer TMD homostructures. To the best of our knowledge, few attempts have been made to achieve difficult engineering of 2D monolayer lateral homosuperlattices (MLHSs), where various arranged phases spontaneously form in a monolayer plane of TMDs. Herein, we demonstrate a MLHS with alternating 1T and 2H phases in a WS₂ single layer. The MLHSs demonstrate

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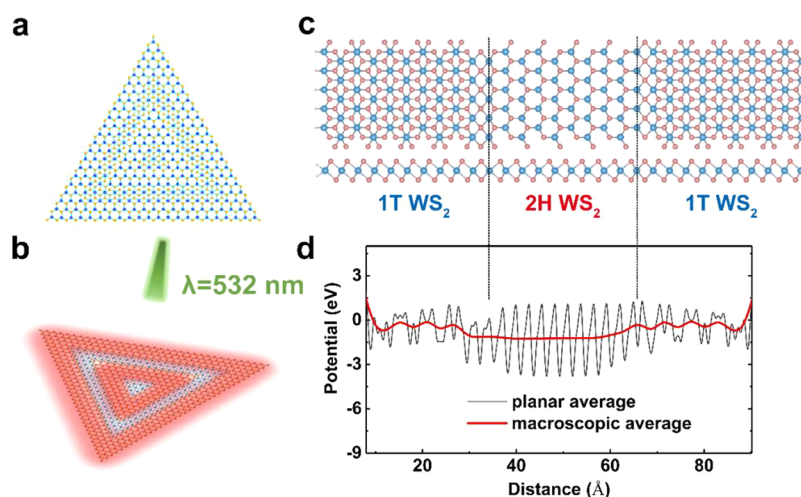


Figure 1. (a) Schematic showing two-dimensional MLHSs with 1T/2H-WS₂. (b) Schematic illustration of 1T/2H-WS₂ MLHSs irradiated with a 532 nm laser. (c) Atomic model and (d) potential distributions of an alternating lateral 1T-2H-1T construction in a WS₂ monolayer simulated with density functional theory.

a lateral periodic distribution of photoluminescence and junction interfaces originating from alternating arrangement of semiconductor phases (2H) and metal phases (1T) in the monolayer plane. The localized electrons in the 2H phases and the surface potential distributions lead to enhanced radiation recombination in the MLHSs with alternating 2H and 1T phases. The emergence of MLHSs in ML-WS₂ will provide a better in-depth understanding of 2D lateral homojunctions and associated optical and optoelectronic behaviors.

RESULTS AND DISCUSSION

Understanding and constructing lateral homojunctions and MLHSs in a monolayer is always a fundamentally challenging task. Figure 1 shows the concept and illustration of MLHSs with an alternating arrangement of the 1T phase and 2H phase in one WS₂ monolayer. Combined with covalent bonds in a WS₂ monolayer plane,^{3,12,20} MLHSs can be regulated within the alternating 1T and 2H phases in a plane, and such construction in a monolayer of TMDs is also rarely explored. An atomic level schematic of MLHSs (Figure 1a) indicates that the trigonal prismatic structure (H-phase) and octahedral prismatic structure (T-phase) can form in TMDs according to the different coordination modes between transition metal atoms and chalcogenide atoms.^{26–28} It is documented that semiconductor 2H-WS₂ exhibits a direct band gap and strong irradiated photoluminescence at room temperature because of the thickness of a single atomic layer.^{29–32} In stark contrast, the metallic 1T phase in WS₂ shows a low work function and low contact resistance.^{26,33,34} Thus, when MLHSs are irradiated with a 532 nm laser, there is a transverse periodic distribution of distinctive photoluminescence in a 2D monolayer, as shown in Figure 1b. In addition, the stronger photoluminescence occurring in the regions of the 2H-WS₂ phase in MLHSs is observed because the electrons are localized in the 2H-phase regions due to the quantum confinement effect and the potential well in MLHSs. Density functional theory (DFT) calculations were performed to further probe the presence of the potential well in MLHSs. There are potential differences between the different phases as known from previous reports.^{35–38} An approximate model of 1T-2H-1T alternating nanoribbons in Figure 1c shows the

formed potential wells in 2D MLHSs (Figure 1d). Thus, the theoretical calculation implies that the alternating 1T and 2H phases can produce a potential well in MLHSs.

In Figure 1, we have shown that the photoluminescence can be alternately distributed in space in a MLHS monolayer, which can be further confirmed by spectroscopy images. Figure 2 shows the optical microscopy photographs and PL intensity maps of various 2D MLHSs. Figure 2a displays a microscopy image of a WS₂ monolayer with another stacked bilayer in the center. We marked the layered WS₂ on a SiO₂/Si substrate with triangular regions using blue dashed lines and the bilayer in the center with a green dashed line. When this WS₂

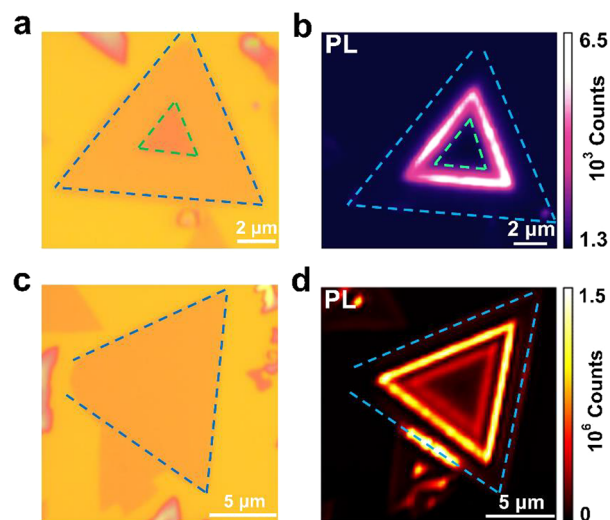


Figure 2. Various 2D MLHS configurations in different WS₂ monolayers. (a) Optical microscopy photographs in monolayer WS₂ with bilayer stacking on a SiO₂/Si substrate. (b) PL intensity mapping of the lateral homosuperlattices (LHSs) in (a). The green dashed triangle region in (a) and (b) is a bilayer and the region between the blue and green dashed triangles is a monolayer, while the highlighted region in (b) is 2H-WS₂. (c) Optical microscopy photographs of a WS₂ monolayer. (d) PL intensity mapping of MLHSs composed of 1T- and 2H-phase WS₂ on a SiO₂/Si substrate.

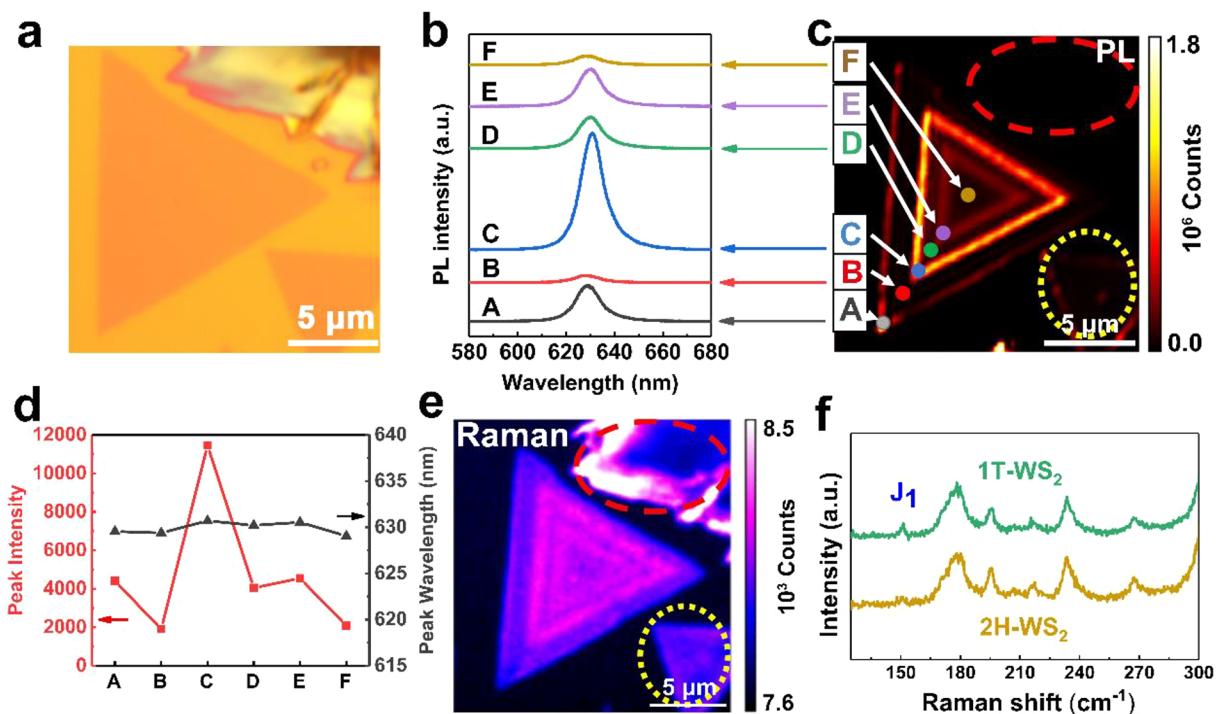


Figure 3. Raman and photoluminescence characterizations of a 2D MLHS with 1T/2H-WS₂ on a SiO₂/Si substrate. (a) Optical microscopy image. (b) Photoluminescence spectra of the A–F positions indicated in (c). (c) PL intensity map. (d) PL peak intensity profile (red) and the PL peak wavelength profile (black) from positions A to F in (b). (e) Raman intensity map corresponding to the PL map in (c). (f) Raman spectra of the 1T phase and 2H phase in monolayer WS₂.

monolayer was irradiated with a 532 nm laser, anomalous PL images were obtained, as shown in Figure 2b. Compared with the regions marked in green (*i.e.*, the stacked bilayer) in Figure 2a,b, we found that the bilayer region hardly exhibits a PL response. Such a phenomenon results from the bilayer.^{39,40} Obvious PL only appears between the blue dashed lines and the green dashed line. Interestingly, the monolayer region shows an arranged PL distribution, which is different from the PL behaviors in monolayer WS₂.^{29,41} As demonstrated in Figure 1, we believe that there is an alternating arrangement of 1T and 2H-WS₂ phases in the monolayer regions in Figure 2a. Moreover, the 2H-phase regions lie between two 1T-phase regions, which leads to such an arranged PL image due to the quantum confinement effect.^{39,42}

To exclude the effect of the varying layer numbers, the WS₂ monolayer was also investigated, as shown in Figure 2c,d. No covered layer was observed in the WS₂ monolayer in the optical microscopy image in Figure 2c, while the PL intensity shows a similar arrangement in Figure 2d. The WS₂ MLHS involves five junctions where the bright regions should be 2H phases and the dark regions should be 1T phases. These results indicate that the triangle is positioned either on top of the superlattice or on the side and does not affect the formation and photoluminescence of such 2D MLHSs. This periodic and localized PL characteristic is dominated by the bilayer stacking, missing edge and edge defects of monolayer WS₂. Both the PL data and the theoretical calculations indicate that only the MLHS configuration results in such a PL arrangement in a single WS₂ monolayer, while ongoing exploration will focus on observing and controlling the microstructures in MLHSs.

To further verify the MLHS with 1T/2H-WS₂, we subsequently utilized Raman spectroscopy and PL spectroscopy to analyze the changes in this WS₂ monolayer. Figure 3a

presents an optical micrograph of 2D MLHS with 1T/2H-WS₂, and the two layers are indistinguishable. It is worth mentioning that there is no observable optical contrast difference or thickness variation (*e.g.*, monolayer and bilayer) in the homosuperlattice, as shown in Figure 2a. This result affirms that WS₂ is homogeneous without a hybrid structure.^{36,43} To delve more into the details, micro-PL spatial mapping is used to further investigate this homosuperlattice structure. Parts b and c of Figure 3 show the characteristic PL spectra and intensity map, respectively. In Figure 3c, the periodic PL image results from the regions of the alternating 1T and 2H phases that exist adjacently in monolayer WS₂. In addition, the corresponding PL spectra imply that there are obvious variations at positions A–F, as shown in Figure 3b. Furthermore, the special periodic localized photoluminescence of the MLHS in the 2D plane and the junction regions of the 1T and 2H phases exhibit abrupt boundaries, as can be seen from Figure 3c. The semiconducting 2H phase has a higher PL intensity, while the 1T phase has no obvious PL, as known from previous reports.¹⁸ Thus, we observed and distinguished the difference between the 1T and 2H phases of WS₂ by the variations in the relative PL intensities in Figure 3c. In our cases, the clear contrast in the PL map provides further evidence of the existence of 1T and 2H phases and the formation of a homosuperlattice. Moreover, PL can provide a way to directly observe 2D MLHS with 1T/2H-WS₂ without any assistance from other measurement techniques. Figure 3d displays the changes in PL intensities at the different regions marked as A–F in Figure 3c. Strong PL intensities (in red) and peak wavelengths (in black) are found in regions A, C, and E, while there are no obvious PL intensities in regions B, D, and F. The red line profile (*i.e.*, PL intensities) shows that the PL intensities alternately change from positions A to F, and the C

position has the strongest PL intensity. Additionally, the black line profile (*i.e.*, PL peak wavelength) does not obviously change and maintains a stable value of approximately 630 nm from points A to F.

In addition to PL mapping, Raman spectra were obtained to check and demonstrate that different phases existed in the WS₂ monolayer with alternating 1T and 2H phases.^{33,36,43,44} Figure 3e shows the Raman intensity map using the WS₂ E_{2g}¹ mode at 350 cm⁻¹, revealing that the 1T and 2H regions had a regular variety of E_{2g}¹ intensities.⁴⁵ It is difficult to distinguish the 1T and 2H phases from the associated WS₂ E_{2g}¹ and A_{1g} position maps (seen in Supporting Information, Figure S1c,d), which insinuates that the flake is homogeneous. As documented elsewhere,^{34,46} the J₁ mode in Figure 3f may be from the 1T phase, which can suggest the difference between the 1T and 2H phases. In addition, the regions marked by the yellow dashed circles show the contrast of the small WS₂ triangle, revealing weak PL emission, which mainly results from the 1T phase. The regions marked by red dashed circles are obvious in the Raman mapping (Figure 3e) but difficult to observe in the PL image (Figure 3c). This result indicates that such fantastic PL emission only occurs in a WS₂ monolayer with alternating 1T and 2H phases although other advanced measurements would also help observe and understand them. However, there are still a few alternating variations between B, D, F and A, C, E, implying that the 1T and 2H phases are positioned in the different domains separately rather than existing as an amalgam in one domain (see Supporting Information, Figure S1a).⁴³

Neutral excitons and negative trions are often exploited to understand the fundamental behavior behind the PL emission in layered WS₂ materials. The behaviors of excitons generally dominate the optical properties and light emission of 2D semiconductor materials.^{31,32,42,43} Figure 4a outlines the

excitons and negative trions were used to subsequently understand the distinctive PL behaviors by the Lorentz fitting of PL spectra.^{45,47–50} In particular, the contribution of trions in the PL in the 2H regions is greater than that in the 1T regions, which also verifies the essential difference between the 2H (semiconductor) phase and 1T (metal) phase.²⁶ In 1T/2H-WS₂ MLHSs, the PL intensity ratio of trions and excitons varies regularly with the alternating 1T and 2H phases, as shown in Figure 4c, implying the existence of completely different electronic states and the formation of MLHSs. Trions typically appear in the 2H phase, and there are more electrons that combine with excitons in that phase than in the 1T phase because of the direct band gap and associated inevitable defects;^{34,47,49–51} thus, an obvious PL emission in the 2H phase is observed.

In addition, we discovered that the binding energy at positions B, D, and F is larger than that at positions A, C, and E when the binding energy of trions, defined as the peak energy difference between the exciton and trion, is calculated (Supporting Information, Figure S2c). Therefore, the charges are confined excitons in the plane, and the exciton binding energy is relatively large.^{47–49} A higher binding energy suggests a higher carrier density in monolayer TMDs, as demonstrated elsewhere.^{26,42,45,48} The higher carrier density is related to the formation of the 1T phase.^{27,28,33,34}

We further used synchronous measurements of the surface morphology and surface potential to identify and understand MLHSs and their optical behaviors. The *in situ* atomic force microscopy (AFM) topography in Figure 5a displays

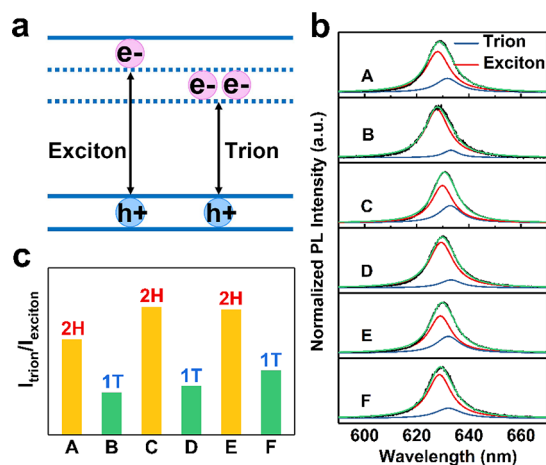


Figure 4. PL profiles of excitons and trions in a 2D MLHS with 1T/2H-WS₂. (a) Schematic diagram of excitons and trions in WS₂. (b) Lorentz fitting for excitons and trions in the PL curves in Figure 3b. (c) PL intensity ratio (trions/excitons) at positions A–F in the MLHSs in Figure 3c.

schematic illustration of the neutral exciton and negative trion in the WS₂ monolayer. The atomic layer thickness causes a strong Coulombic interaction between electrons and holes in the WS₂ monolayer.^{12,30} Both neutral excitons and charged excitons exist with a high exciton density in monolayer WS₂.⁴⁷ Figure 4b further shows the Lorentz fitting of PL spectra at the positions marked as A–F in Figure 3c. Two peaks for neutral

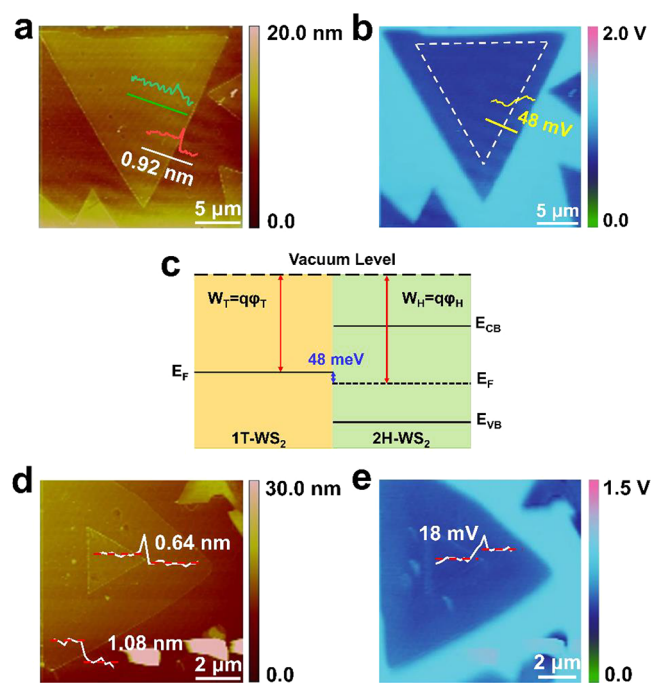


Figure 5. Surface morphology (AFM) and surface potential (KPFM) characterizations of 2D MLHSs with 1T/2H-WS₂. (a) Synchronous AFM and (b) KPFM images of the MLHS in Figure 3a. The surface potential interface is marked by a white dashed triangle. The contact potential difference (CPD) line profile is marked with a yellow line across the 2H-1T interface. (c) Energy band diagram for the 1T/2H-WS₂ superlattices in Figure 3a. (d) Synchronous AFM and (e) KPFM images of the 2D LHS in Figure 2a.

monolayer WS₂ with a thickness of approximately 0.92 nm according to the white height line profile. In addition, the uniform height profile in the green line further reveals the formation of seamless 2D MLHSs.^{30,33,41,52} Along the white dashed triangle, we observe that there is an obvious contact potential difference (V_{CPD}) or work function difference between the 1T and 2H phases in the corresponding Kelvin probe force microscopy (KPFM) image (Figure 5b). The surface potential of the 1T region is higher, and the surface potential difference is approximately 48 mV across the interface of the 1T and 2H domains, which is in congruence with the V_{CPD} reported in other literature.^{32,36,43} Hence, charge transfer might occur from 1T regions to 2H regions, suggesting electronic doping. More electrons in the 2H regions are localized in the potential well formed by the MLHS structure with 1T/2H-WS₂. Thus, it is more likely to form trions and compound emission in the well.^{45,48,49,53} Therefore, the PL emission is preferably produced in the 2H regions of MLHSs, while the PL intensity is higher than that previously observed in pure 2H-WS₂.^{30,41,43} All of these results are in synchrony with the measurements and analyses above.

Figure 5c shows the energy band diagram for the 1T/2H-WS₂ lateral superlattice. The work function in the KPFM characterization is sensitive to the Fermi level of 2D materials.⁵⁴ Regarding the monolayer WS₂ with a stacked bilayer region in Figure 5d,e, the surface potential difference is merely approximately 18 mV between the WS₂ monolayer and bilayer. The AFM and KPFM characterizations further demonstrate that the periodic PL in monolayer WS₂ originates from MLHSs with alternating 1T and 2H phases and is independent of layer thickness or vertical stacking (Figure 5d,e). These investigations herald an exceptional vista of understanding and constructing MLHSs with anomalous optical and electronic behaviors. The analysis of PL behaviors demonstrates a simple route for further insights into the relationship between crystal structure and optical properties.^{10,16}

CONCLUSIONS

We demonstrated MLHSs with alternating 1T and 2H domains in a WS₂ monolayer on a SiO₂/Si substrate. Periodic and localized PL emissions were explored through Raman spectroscopy, photoluminescence, KPFM characterization, and DFT calculations. The charge transport from the 1T to 2H regions, and their subsequent localization by the potential wells in MLHSs is responsible for the distinctive PL in 2H regions. While the PL dynamics and formation mechanism of MLHSs with 1T and 2H phases need more exploration, our studies create more opportunities for better understanding the phase engineering of a monolayer plane and for modulating the optical or electronic behavior of 2D materials in next-generation nanoscale optoelectronic devices.

METHODS

MLHSs were fabricated on SiO₂/Si substrates by using powdered WO₃ (99.9% purity, Sigma-Aldrich) and S (99.9% purity, Sigma-Aldrich) sources and adjusting the source temperature during a vapor–solid growth process. All procedures were based on the previous report.⁵⁴ Briefly, AFM and KPFM images were obtained on Veeco/DI multimode SPM, while optical microscopy was performed on Leica DM4000M. Raman and PL spectra and mapping were recorded on a Nanofinder 30 instrument (TII Tokyo Instruments, Inc.). An 1800 g/mm grating for Raman and a 300 g/mm grating for

PL were used, while the measurements were performed with a 532 nm laser (0.02 mW and 0.015 mW) at room temperature.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsnano.1c07803>.

Raman spectra and PL for the 1T and 2H phases of WS₂; Raman position map of the E_{2g}¹ and A_{1g} phonon modes; variation of excitons and trions with alternating 1T and 2H phases according to the Lorentz fitting curves (PDF)

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Notes

The authors declare no competing financial interest.

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